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The effect of swelling on the longitudinal acoustic mode in crystalline α,ω -methoxy-poly(ethylene oxide)

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Low molecular weight poly(ethylene oxide) crystallizes into well defined layer structures. Raman scattering from the longitudinal acoustic mode (LAM) of vibration of the lamellae is readily detected¹⁻³. The experimental evidence¹⁻⁶ favours a structure in which crystalline and non-crystalline layers alternate. It is possible to swell the non-crystalline layers with low molecular weight liquids, and oligomers of ethylene oxide have been used for this purpose^{2,3,7}. For α,ω -hydroxy-poly(ethylene oxide) of molecular weight 2000 g mol⁻¹ swollen with α,ω -hydroxy-oligomers of various molecular weights less than $\bar{M}_n = 600$ g mol⁻¹, it is found^{2,3} that the LAM frequencies vary systematically with the lamella spacing (l_x = thickness of crystalline plus non-crystalline layer) determined by small-angle X-ray scattering (SAXS). These results are given in *Figures 1(a)* and *2(a)*, where the hydroxy ended samples are denoted by molecular weight and suffix H. We plot $\nu_1 l_x$ and ν_3/ν_1 against l_x (ν_1 is the frequency of the LAM fundamental and ν_3 that of the third overtone). For the ideal case of crystalline lamellae vibrating independently of the non-crystalline layer, $\nu_1 l_x$ would increase with l_x and ν_3/ν_1 would be constant and equal to 3. Our results show a considerable effect of the non-crystalline layer on the LAM.

For the α,ω -hydroxy systems the effect of swelling is independent of the molecular weights of the oligomers studied ($\bar{M} = 106$ to $\bar{M}_n = 600$: see *Figures 1* and *2*). However the liquid properties of α,ω -hydroxy-poly(ethylene oxide) may be insensitive to molecular weight. For example^{8,9} the liquid density of α,ω -hydroxy-oligomers of ethylene oxide is constant at 1.12 g cm⁻³ (20°C). This is in contrast to the liquid densities of α,ω -methoxy-oligomers which increase from 0.94 g cm⁻³ ($\bar{M} = 134$ g mol⁻¹) to 1.08 g cm⁻³ ($\bar{M}_n = 600$ g mol⁻¹) and to 1.12 g cm⁻³ only in the high molecular weight limit of extrapolation. Hydrogen bonding, principally of hydroxy to ether oxygen, provides a rationalization of this effect. Consideration of this point prompted us to reinvestigate the effect of molecular weight of the swelling agent on the LAM under conditions where hydrogen bonding is absent.

Samples of α,ω -hydroxy-poly(ethylene oxide) of molecular weights $\bar{M}_n = 2000, 600$ and 200 g mol⁻¹, obtained from various commercial sources, were methoxylated by the procedure described elsewhere⁹. Methoxylated samples are denoted by the original molecular weight with suffix M. Conversion of hydroxy to methoxy was better than 98%. Molecular weight distributions were checked by gel permeation chromatography and were essentially unchanged by the methoxylation process. The crystallizable α,ω -methoxy-poly(ethylene oxide) 2000M had a narrow molecular weight distribution, $\bar{M}_w/\bar{M}_n = 1.03$. Dimethyldigol ($\bar{M} = 134$ g mol⁻¹) was Synchemica grade (Hopkins and Williams) used without further purification. Mixtures were prepared by melting at 70°C and shaking several times over a period of 30 min. Crystallization was at 25°C. LAM frequencies and lamella spacings were determined by laser Raman spec-

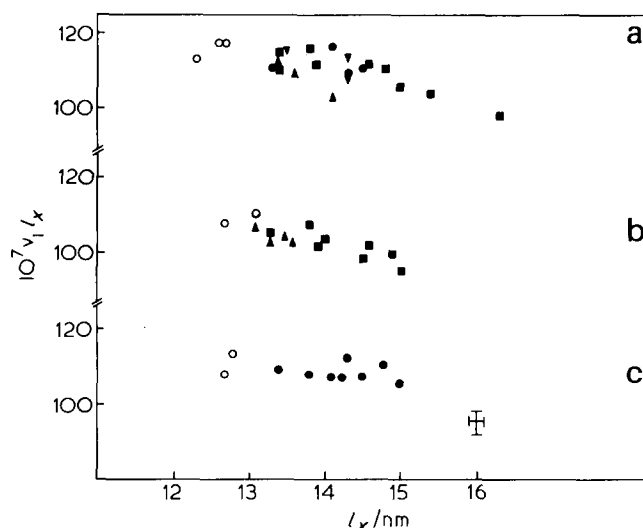


Figure 1 The variation of $\nu_1 l_x$ with l_x for mixtures of: (a) 2000H (○) with 106H (●), 200H (■), 400H (▼) and 600H (▲); (b) 2000M (○) with 200M (■) and 600M (▲); (c) 2000M (○) with 134M (●). The experimental error is indicated

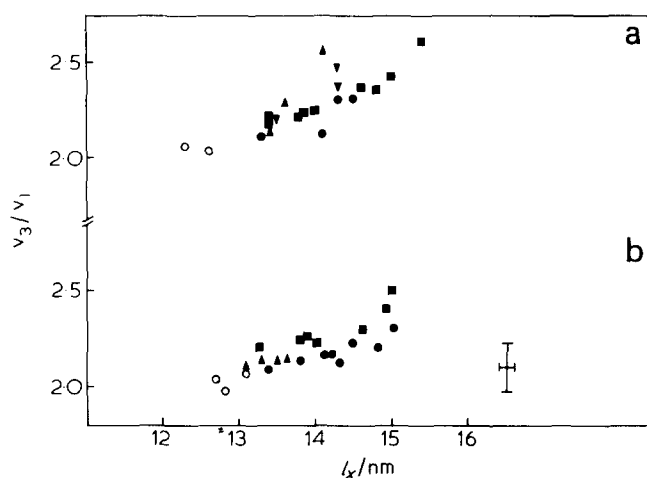


Figure 2 The variation of ν_3/ν_1 with l_x for mixtures of: (a) 2000H (○) with 106H (●), 200H (■), 400H (▼) and 600H (▲); (b) 2000M (○) with 134M (●), 200M (■) and 600M (▲). The experimental error is indicated

Table 1 Lamella spacing (l_x) and LAM fundamental ($\bar{\nu}_1$) and third overtone ($\bar{\nu}_3$) for poly(ethylene oxide) 2000

Sample	α,ω -hydroxy			α,ω -methoxy		
	l_x/nm	$\bar{\nu}_1/\text{cm}^{-1}$	$\bar{\nu}_3/\text{cm}^{-1}$	l_x/nm	$\bar{\nu}_1/\text{cm}^{-1}$	$\bar{\nu}_3/\text{cm}^{-1}$
1	12.6	9.3	19.0	13.1	8.4	19.0
2	12.3	9.2	19.0	12.8	8.8	19.0
3	12.3	9.2	19.0	12.7	8.5	19.0

trosopy and small-angle X-ray scattering as described earlier^{1,2}.

The experimental results are given in Tables 1 and 2. The estimated experimental errors are ± 0.2 nm (l_x), ± 0.2 cm^{-1} ($\bar{\nu}_1$) and ± 1 cm^{-1} ($\bar{\nu}_3$). We have investigated 3 different samples of 2000M: one set of results for 2000M (1)/200M has been published earlier². Methoxylation has the effect of increasing l_x by about 0.5 nm and decreasing $\bar{\nu}_1$ by about 0.6 cm^{-1} in comparison with the α,ω -hydroxy precursor (see Table 1). In Figures 1(b,c) and 2(b) the product $\nu_1 l_x$ and the quotient ν_3/ν_1 are plotted against l_x . The results for 2000M/200M and 2000M/600M are indistinguishable and are comparable in most respects with those found for the α,ω -hydroxy systems. The variation of $\nu_1 l_x$ with l_x (but not ν_3/ν_1 with l_x) is less marked for 2000M/134M than for the other systems. We infer that swelling with oligomers of molecular weight 200 g mol^{-1} or more suffices, in this experiment, to model the limit of high polymer in the non-crystalline layer.

We have used^{2,3} the composite rod model with free ends¹⁰ to rationalize our results for α,ω -hydroxy-poly(ethylene oxide). Clearly the same model can be used for α,ω -methoxy-poly(ethylene oxide). The fit to the data² requires a degree of crystallinity of the crystallizable poly(ethylene oxide) 2000 of 70% and an elastic modulus ratio (E_a/E_c) near 0.1. However the crystalline elastic modulus needed for the fit is $E_c \approx 9 \times 10^{10}$ $\text{N } \frac{1}{2}\text{m}^{-2}$ which is much larger than the value of 10^{10} $\text{N } \frac{1}{2}\text{m}^{-2}$ found by lattice extension measurements¹¹. Recent theoretical work of Hsu *et al.*¹² permits prediction of the LAM frequencies of composite rods with perturbing forces at the ends. A uniform rod of elastic modulus 1.0×10^{10} $\text{N } \frac{1}{2}\text{m}^{-2}$ with fixed ends (very large forces) would have $\nu_1 l_x = 96 \times 10^{-7}$ ($\rho = 1.21 \times 10^3$ kg m^{-3}) compared to an

Table 2 Lamella spacing (l_x) and LAM fundamental ($\bar{\nu}_1$) and third overtone ($\bar{\nu}_3$) for α,ω -methoxy-poly(ethylene oxide) 2000 swollen by different weight fractions of low molecular weight oligomers

Weight fraction	l_x/nm	$\bar{\nu}_1/\text{cm}^{-1}$	$\bar{\nu}_3/\text{cm}^{-1}$
2000M(2)/134M			
0.08	14.3	7.8	18.0
0.14	14.5	7.4	18.0
0.18	14.8	7.4	17.8
0.20	15.0	7.0	17.5
2000M(3)/134M			
0.07	13.4	8.1	18.5
0.12	13.8	7.8	18.2
0.16	14.1	7.6	18.0
0.19	14.2	7.5	17.8
2000M(3)/200M			
0.04	13.3	7.9	19.0
0.09	14.0	7.4	18.0
0.15	14.6	7.0	17.5
0.20	14.9	6.7	17.5
2000M(3)/600M			
0.06	13.1	8.2	18.4
0.12	13.3	7.9	18.4
0.18	13.5	7.7	18.0
0.22	13.6	7.6	17.8

experimental value of about 110×10^{-7} . It is not obvious why a rod with fixed ends should model the behaviour of our polymers. It may be that adjacent lamellae in low molecular weight poly(ethylene oxide) are coupled¹², though this is not found for other systems¹³. The possibility of a detailed theoretical¹² fit of our results, incorporating an acceptable value of E_c , is under consideration.

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